# Mathematical model of electromigration-driven evolution of the surface morphology and composition for a bi-component solid film

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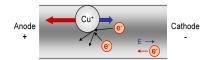
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# **Introduction: Surface electromigration**

#### Flat film

Figure : Single-crystal, metallic film grown on a substrate by MBE or CVD.

**Surface electromigration:** *Drift* of adatoms on **heated crystal surfaces** of metals in response to applied DC current, due to the *momentum transfer* from electrons to adatoms through scattering ↔ "Electron wind"



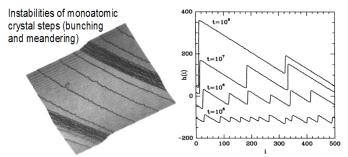
Adatom = mobile, adsorbed atom or ion

### Introduction: Example: Current-driven surface faceting of surfaces

Continuum theory of surface dynamics on length scales much larger than the interstep distance: Krug & Dobbs'1994, Schimschak & Krug'1997

If the electric field is horizontal (along planar, unperturbed surface), then:

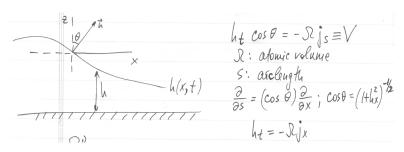
- 1. Facet orientations are first established locally,
- 2. Further evolution proceeds through a coarsening process



Facet size (hill-to-hill distance)  $X \sim t^n$ , where  $n \approx 0.25$ Each facet = many monoatomic crystal steps



# Introduction: Morphological evolution of a single-component surface (W.W. Mullins'57-65)



$$j = -\frac{\nu}{kT}DM(\theta)\left(\mu_s + qE\cos\theta\right)$$
: adatoms flux on the film surface when **E** is along the *x*-axis

 $DM(\theta)$ : anisotropic diffusivity of the adatoms (surface orientation-dependent), where  $M(\theta)$  is the *mobility* 

 $\mu(s) = \Omega \gamma \kappa$ : the chemical potential (Mullins'57)



# The model for a bi-component surface

$$h_t = V/\cos\theta, \ \ V = -\Omega\left(\frac{\partial J_A}{\partial s} + \frac{\partial J_B}{\partial s}\right),$$

where  $J_A$  and  $J_B$  are the surface diffusion fluxes of the components A and B:

$$J_{i} = -\frac{\nu D_{i}}{kT} M_{i}(\theta) C_{i}(s, t) \left[ \frac{\partial \mu_{i}}{\partial s} + qE \cos \theta \right], \ i = A, B$$

Here  $C_A(x,t)$  and  $C_B(x,t)$  are the dimensionless surface concentrations of adatoms A and B, defined as products of a volumetric number densities and the atomic volume. Then,

$$C_A(x,t)+C_B(x,t)=1.$$

$$\mu_i = \Omega \gamma_i \kappa + k T ln \frac{C_i}{1-C_i} \approx \Omega \gamma_i \kappa + -2kT + 4kTC_i,$$

when the "mixture" contribution is linearized about  $C_i = 1/2$ .

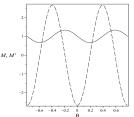
This form of the "mixture" contribution implies a thermodynamically stable alloy, thus the natural surface diffusion acts to smooth out any compositional nonuniformities. On the other hand, the electromigration may be the cause of their emergence and development.

# The model for a bi-component surface, continued

The anisotropic adatom mobility (Schimschak & Krug'1997):

$$M_i(\theta) = rac{1 + eta_i \cos^2\left[N_i(\theta + \phi_i)
ight]}{1 + eta_i \cos^2\left[N_i\phi_i
ight]}, \quad ext{where}$$

 $N_i$ : the number of symmetry axes,  $\phi_i$ : the angle between a symmetry direction and the average surface orientation,  $\beta_i$ : anisotropy strength.



PDE governing evolution of the surface concentration  $C_B(s,t)$  (Spencer, Voorhees and Tersoff'93):

$$\delta \frac{\partial C_B}{\partial t} + \mathbf{C_B} \mathbf{V} = -\Omega \frac{\partial J_B}{\partial s},$$

where  $\delta$  is the thickness of the surface layer and quantifies the "coverage".



The dimensionless problem (using  $[x] = h_0, [t] = h_0^2/D_B$ )

$$\begin{aligned} \mathbf{h_{t}} = & \frac{4}{mQ} \frac{\partial}{\partial x} \left\{ \left( 1 + h_{x}^{2} \right)^{-1/2} \left[ DM_{A} \left( h_{x} \right) \left( 1 - C_{B} \right) \left( R_{A} \frac{\partial \kappa}{\partial x} - \frac{\partial C_{B}}{\partial x} + F \right) + \right. \\ & \left. M_{B} \left( h_{x} \right) C_{B} \left( R_{B} \frac{\partial \kappa}{\partial x} + \frac{\partial C_{B}}{\partial x} + F \right) \right] \right\}, \\ & \left. \frac{\partial \mathbf{C_{B}}}{\partial \mathbf{t}} = - \left( 1 + h_{x}^{2} \right)^{-1/2} \left[ \mathbf{QC_{B} h_{t}} - \frac{4}{m} \right. \\ & \left. \frac{\partial}{\partial x} \left\{ \left( 1 + h_{x}^{2} \right)^{-1/2} M_{B} \left( h_{x} \right) C_{B} \left( R_{B} \frac{\partial \kappa}{\partial x} + \frac{\partial C_{B}}{\partial x} + F \right) \right\} \right]. \end{aligned}$$

Here the parameters are:

$$R_i = \frac{\Omega \gamma_i}{4kTh_0}, \ F = \frac{\alpha \Delta Vq}{4nkT}, \ Q = \frac{h_0}{m\Omega \nu}, \ D = \frac{D_A}{D_B}.$$

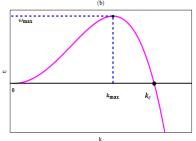
This assumes:

- $E = \Delta V/L$ , where  $\Delta V$  is the applied potential difference and  $L = nh_0$  is the lateral dimension of the film (n > 0 is a parameter),
- $\delta=m\Omega\nu$ , where m>0 is a parameter;  $\Omega\nu$  is monolayer thickness



# Linear stability analysis (LSA), general idea

- Perturb the surface about the equilibrium constant height  $h_0$  by the perturbation  $\xi(x,t)$ , obtain PDE for  $\xi$
- (ii) Linearize PDE for  $\xi$  and obtain linear PDE  $\xi_t = F(\xi, \xi_x, \xi_{xx}, ...)$
- (iii) Take  $\xi = e^{\omega t} \cos kx$  and substitute in PDE  $\rightarrow \omega(h_0, k, R_i, n, m, D)$
- (iv) Examine for what values of the parameters the growth rate is positive or negative.  $\omega < 0$ : surface is stable;  $\omega > 0$ : surface is unstable



Long-wave instability.  $k_c$ : the cut-off wavenumber;  $k_{max}$ : a wavenumber detected in experiment 4□ > 4同 > 4 = > 4 = > ■ 900

- $h(x,t)=1+\xi(x,t),\ C_B(x,t)=C_B^0+\hat{C}_B(x,t)$ : the perturbations
- $\xi(x,t) = Ue^{\omega(k)t}e^{ikx}$ ,  $\hat{C}_B(x,t) = Ve^{\omega(k)t}e^{ikx}$ , where U, V are (unknown) constant and *real-valued* amplitudes
- $\omega = \omega^{(r)}(k) + i\omega^{(i)}(k)$
- Also expand:  $M_i(h_x) = M_i(0) + M_i'(0)h_x$ , where  $M_i(0)$ ,  $M_i'(0)$  are the parameters

This results in the quadratic eqn. for  $\omega^{(r)}(k)$ :

$$\omega^{(r)}(k)^{2} + \omega^{(r)}(k) \left[ k^{2} C_{B}^{0} \left( M_{B}(0) \left( 1 - \frac{m}{4} C_{B}^{0} \right) + \frac{M_{A}(0)}{4} mD \left( 1 - C_{B}^{0} \right) \right) + \frac{k^{2}}{Q} \left\{ DFM'_{A}(0) \left( 1 - C_{B}^{0} \right) + FM'_{B}(0) C_{B}^{0} + k^{2} DR_{A} M_{A}(0) \left( 1 - C_{B}^{0} \right) + k^{2} R_{B} M_{B}(0) C_{B}^{0} \right\} \right] + \frac{k^{4}}{Q} \left[ DF \left( M_{B}(0) M'_{A}(0) + M_{A}(0) M'_{B}(0) \right) C_{B}^{0} \left( 1 - C_{B}^{0} \right) + k^{2} D \left( R_{A} + R_{B} \right) M_{A}(0) M_{B}(0) \left( 1 - C_{B}^{0} \right) \right] = 0,$$

#### LSA, continued

And,

$$\omega^{(i)}(k) = \left[M_B(0)\left(1 - \frac{mC_B^0}{4}\right) + \frac{M_A(0)}{4}DmC_B^0\right]kF.$$

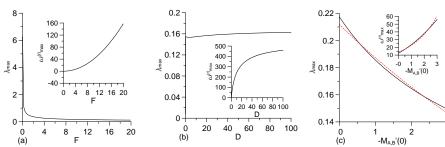
Thus the perturbations experience the lateral drift with the speed  $v = |\omega^{(i)}(k)/k| \sim F$ , which does not depend on k.

# **LSA**, continued: analysis of $\omega^{(r)}(k)$

- The limit of vanishing surface layer thickness: As  $\delta \to 0$   $(Q \to \infty)$ ,  $\omega^{(r)}(k) < 0$
- For finite Q, the longwave instability with

$$k_c = \left[ F \left( 1 - C_B^0 \right) \frac{M_B(0) M_A'(0) + M_A(0) M_B'(0)}{M_A(0) M_B(0) \left( C_B^0 - 1 \right) \left( R_A + R_B \right)} \right]^{1/2} \sim F^{1/2}$$

Note that  $k_{max} \neq k_c/\sqrt{2}$ ; obtain  $\lambda_{max} = 2\pi/k_{max}$  numerically

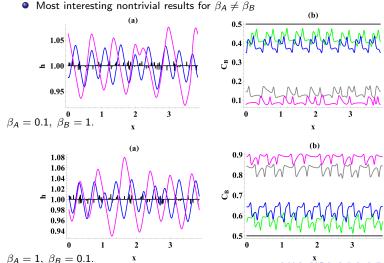


$$\lambda_{max} \sim F^{-1/2}$$
,  $\omega_{max}^{(r)} \sim F^2$ ; Dashed lines in (c): fits

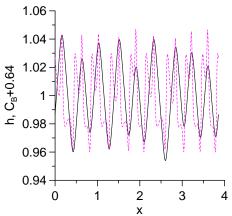
$$\lambda_{max} = 0.212 \exp(0.123 M'_{A,B}(0)), \ \omega_{max}^{(r)} = 13.8 \exp(-0.493 M'_{A,B}(0)).$$

# **Nonlinear surface dynamics**

- Comp. domain  $0 \le x \le 20\lambda_{max}$ , initial condition:  $h(x,0) = 1 + \text{small random perturbation}, C_B(x,0) = 1/2$ , periodic b.c.'s
- Perpetual coarsening  $\rightarrow$  hill-and-valey structure  $\rightarrow$  scaling law ?



# Nonlinear surface dynamics, continued

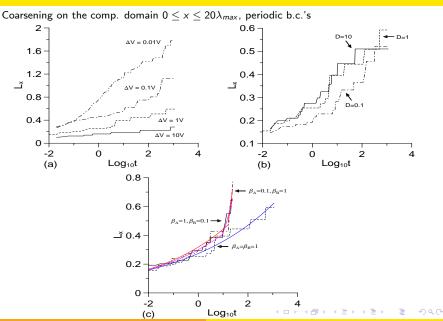


Concentration of B adatoms (dashed line) superposed over the corresponding surface shape (solid line).  $\beta_A=0.1,\ \beta_B=1.$ 

Let  $L_x$  be the mean distance between neighbor valleys



# Nonlinear surface dynamics, continued



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Electromigration-driven evolution of the surface morphology and

# Nonlinear surface dynamics, continued

# Coarsening laws:

vs. ΔV:

$$\begin{array}{l} L_x^{(\Delta V=0.01V)} = 0.77 t^{0.138}, \ L_x^{(\Delta V=0.1V)} = 0.46 t^{0.126}, \\ L_x^{(\Delta V=1V)} = 0.273 t^{0.118}, \ L_x^{(\Delta V=10V)} = 0.156 t^{0.074}, \end{array}$$

• vs. *D*:

$$L_x^{(D=0.1)} = 0.205t^{0.138}, \ L_x^{(D=1)} = 0.273t^{0.118}, \ L_x^{(D=10)} = 0.29t^{0.104}$$

• For  $\beta_A = 0.1, \ \beta_B = 1$ :

$$L_{\rm x}^{(eta_A=0.1,eta_B=1)}=0.318t^{0.143} \ {
m for} \ 0 \leq Log_{10}t \leq 1.22, \ L_{\rm x}^{(eta_A=0.1,eta_B=1)}=0.04t^{0.905} \ {
m for} \ 1.22 \leq Log_{10}t \leq 1.38$$

• For  $\beta_A = 1$ ,  $\beta_B = 0.1$ :

$$L_x^{(eta_A=1,eta_B=0.1)}=0.3t^{0.135} \ {
m for} \ 0 \leq Log_{10}t \leq 1, \ L_x^{(eta_A=1,eta_B=0.1)}=0.128t^{0.514} \ {
m for} \ 1 \leq Log_{10}t \leq 1.38$$



# **Summary**

- Found the long-wavelength instability coupled to the lateral surface drift. The drift is not present in the similar model for the one-component film
- The perturbations wavelength  $\lambda_{max}$  and the growth rate  $\omega_{max}^{(r)}$  scale as  $F^{-1/2}$  and  $F^2$ , respectively, where F is the applied electric field parameter
- ullet  $\lambda_{ extit{max}}$  sharply decreases when the derivatives of the diffusional mobilities increase
- If  $\beta_A$  differs significantly from  $\beta_B$ , the surface is enriched by one atomic component. The second component is absorbed into the solid
- ullet Increase of  $\Delta V$  makes the surface rougher (coarsening slows down)
- If  $\beta_A$  differs significantly from  $\beta_B$ , then there is a pronounced speed-up of coarsening at the late times (a factor of 4-6 increase in the coarsening exponent)

#### THANKS !1